

Highly Photostable Perovskite Nanocubes: Toward Integrated Single Photon Sources Based on Tapered Nanofibers

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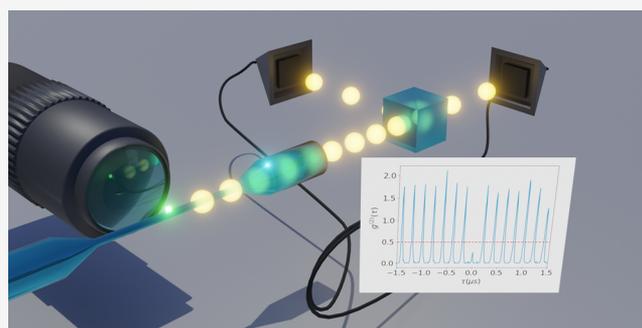
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Supporting Information

ABSTRACT: The interest in perovskite nanocrystals (NCs) such as CsPbBr₃ for quantum applications is rapidly raising, as it has been demonstrated that they can behave as very efficient single photon emitters. The main problem to tackle in this context is their photostability under optical excitation. In this article, we present a full analysis of the optical and quantum properties of highly efficient perovskite nanocubes synthesized with an established method, which is used for the first time to produce quantum emitters and is shown to ensure increased photostability. These emitters exhibit reduced blinking together with a strong photon antibunching. Remarkably these features are hardly affected by the increase of the excitation intensity well above the emission saturation levels. Finally, we achieve for the first time the coupling of a single perovskite nanocube with a tapered optical nanofiber in order to aim for a compact integrated single photon source for future applications.

KEYWORDS: perovskites, single photon sources, quantum dots, nanocrystals, nanofibers



The interest in perovskites, originally studied for solar-cell applications,¹ has recently increased in the quantum optics community. Perovskite nanocrystals are indeed versatile emitters and the possibility to tune their emission wavelength playing on their size and composition, together with their coherent emission² and the ability to obtain single photon emission at low^{3,4} and room⁵ temperatures makes them promising nano-objects for quantum applications. Interestingly, the appropriate choice of the perovskite composition allows the achievement of efficient emission in the near-infrared range,⁶ which is very hard to obtain with II–VI colloidal nanocrystals. Moreover, they are easily synthesized by low-cost, well-mastered wet-chemistry techniques, differently from single defects in nanodiamonds and epitaxial quantum dots, which require heavy fabrication facilities.

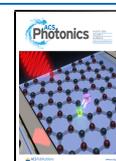
Despite these strong points,⁷ the optical stability is still the main limitation in their use as they usually bleach after a few minutes under illumination. Multiple approaches have been attempted to reduce this problem such as polymer encapsulation,^{8,9} but also alumina encapsulation using atomic layer deposition^{10,11} and surface passivation.¹² Although these methods have shown some effects in reducing the bleaching, only partial results are achieved. Indeed, the first technique is not suitable for applications such as the coupling of single emitters with photonic devices, as it requires the NCs to be surrounded by a dense polymer matrix. The second technique

is more promising; however, no single photon emission has been demonstrated so far using this method.

We report here a method of fabrication,¹³ used for the first time for this purpose, that allows us to obtain higher stability samples of perovskite nanocrystals which can be excited under optical excitation for more than 1 h. We also investigate the role of the dilution on the stability, suggesting new approaches to address this problem. Thanks to this improved stability, we were able to perform a full characterization of the optical and quantum features of perovskite nanocrystals, showing at the same time reduced blinking and strong photon antibunching of the emission. Finally, for the first time with such emitters, we achieve the coupling of a single perovskite nanocrystal with a tapered optical nanofiber. As it has been shown with atoms^{14,15} and solid state emitters,^{16–19} this technique is of paramount importance for applications in the emerging field of quantum technologies. Our result is a promising step toward the realization of a compact integrated single photon device at room temperature with perovskite nanoemitters.

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PEROVSKITE NANOCUBE FABRICATION

In this work, we target particles with single photon emission properties. To reach this goal, confinement is required to obtain a two level system-like emitter. In addition the particle needs to be sufficiently colloidally stable to avoid aggregation. Because of the carrier confinement requirement, bulk-like conventional nanocrystals obtained from the method described in the work of Protesescu et al.²⁰ cannot be used. Moreover, the obtained cubes (see Figure S1) presents a limited colloidal stability under dilute conditions due to the fact that the alkyl ammonium ligands have strong binding dynamics²¹ to the nanocrystals surface. It was recently proposed to use alternative ligands such as zwitterion²² or phosphonic acid²³ to enhance the binding of the ligands and overcome the loss of stability under dilution.

Alternatively, the confined form of perovskite material can also be obtained from nanoplatelets; however in this case, the large lateral extension of the obtained objects also leads to poor colloidal stability. This is why we chose to target small confined nanocubes with a particle shape combining quantum confinement together with a reasonable colloidal stability. The procedure we use has been initially developed for the growth of CsPbBr₃ nanosheets.¹³

Compared to the work of Protesescu et al.²⁰ there are three major changes:

1. Less cesium oleate is introduced to favor the growth of a Cs free phase.
2. Two additional ligands with saturated alkyl chains are introduced (octanoic acid and octyl amine) to favor the crystallization of the Cs free phase.
3. The reaction time is extended from 30 s to 35 min to favor the growth step.

In this synthesis, the main product is CsPbBr₃ nanoplatelets²⁴ and presents an absorption peak at 430 nm as shown in Figures 1a and S2 for TEM. Moreover, there are two additional

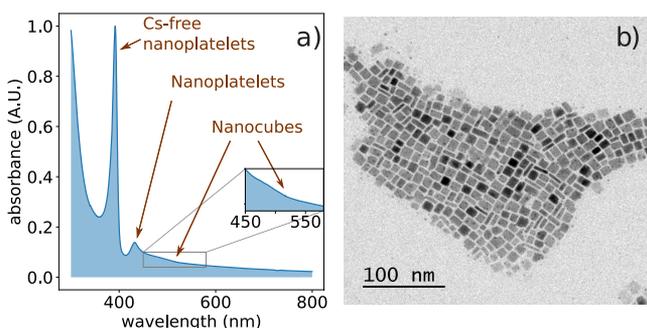


Figure 1. (a) Absorption spectrum of the products obtained from the synthesis with the method described in the text before the last centrifugation. The contribution of the three phases is evidenced. (inset) Portion of the spectrum corresponding to nanocubes contribution. (b) Transmission electron microscopy image of CsPbBr₃ nanocubes responsible for the photoluminescence.

products, the first one is made of Cs-free nanoplatelets where a plane of lead bromide is sandwiched between two planes of ligands with C 8 chains. This phase presents a clear peak at 398 nm in the absorption spectrum.²⁴ The second one is composed of synthesized cubes responsible for the small absorption edge from 450 to 500 nm. This phase is actually the one in which we are interested. We can see the signature of these confined

cubes as a small change in the slope of the absorption curve, visible in the inset of Figure 1a.

These small cubes are far smaller than the nanoplatelets and they can be easily sorted thanks to selective precipitation, as shown in Figure S2. TEM imaging reveals a parallelepipedic or nanoplatelet-like shape with small aspect ratio (3–4) for these particles; see Figures 1b and S3. As detailed in the Supporting Information, the average thickness has been determined to be 3.6 nm which is below the Bohr radius and induces confinement, while the average lateral extension is found to be 11.3 nm, leading to a lack of confinement in the plane. The presence of confinement is confirmed by the blue shift of the PL from these particles compared to the ones obtained for the large cubes of the Protesescu's method. The obtained CsPbBr₃ nanocubes appear to be more stable upon dilution than the ones obtained by the direct procedure, as shown in the following.

Optical Characterization. We optically characterized the nanocrystals (NCs) using an inverted confocal microscope: the experimental setup is shown in Figure 2a.

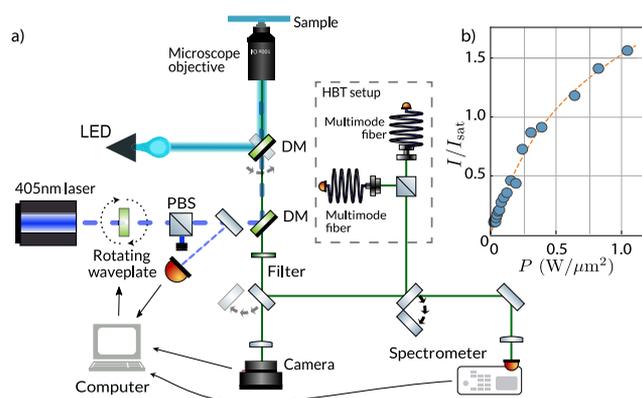


Figure 2. (a) Inverted confocal microscope setup for the sample analysis. The emitters can be excited with a LED or a pulsed laser. The emission can be detected on a camera or analyzed with a spectrometer. Finally, we can perform a $g^{(2)}$ measurement with a Hanbury Brown and Twiss setup (HBT setup), with multimode optical fibers connected to APDs. DM: dichroic mirror. PBS: polarizing beam splitter. (b) Saturation measurement of a single NC. The blue dots are the experimental data while the orange line is the fitted function from eq 1.

The samples are prepared for optical characterization measurements by spin-coating the colloidal solution on a glass coverslip. A LED lamp at 400 nm is first used to locate the emitters and an image of the field of view of the microscope is collected by a CMOS camera. A single emitter is then excited via a 405 nm picosecond pulsed laser (pulse width <50 ps) with a repetition rate adjustable from 2.5 to 5 MHz. After filtering out the excitation with a 430 nm long-pass filter, the luminescence of a single perovskite nanocube is collected by an inverted confocal microscope and sent to the optical characterization part. All the measurements were performed at room temperature.

A typical emission spectrum is reported in Figure 3a and shows a central wavelength of 500 nm with a full width at half-maximum (fwhm) of about 15 nm. The distribution of the emitted wavelengths with the corresponding fwhm is shown in Figure 3b for 78 emitters.

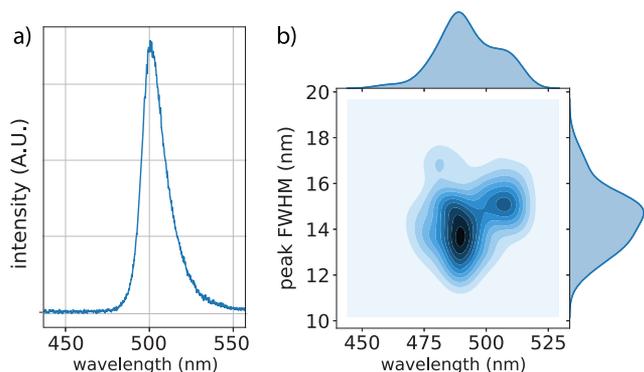


Figure 3. (a) Typical emission spectrum of a single NCs and (b) central wavelength and fwhm distributions of the emission for a sample of 24 emitters.

Here, we clearly see two main peaks with central emission wavelength (CEW) at 485 and 510 nm, respectively. The mean fwhm of the emitted light is about 14 nm for the first peak and 15 nm for the second one. By comparison with the bulk perovskite emission wavelength at about 520 nm,^{25–27} we attribute the former emission to small single photon emitters NCs with slight quantum confinement and the latter emission to large size nanocubes for which the confinement is mostly absent. As shown in the following, the small size nanocubes exhibit strongly antibunched emission, highlighting for the first time in this kind of emitters the crucial role of the charge confinement on their quantum properties. We report a study over 24 different emitters. For each emitter we measured the emitted power as a function of the excitation intensity. The background is subtracted from the experimental data. By fitting the data using the following saturation function, we are able to extract the saturation intensity I_{sat} :

$$P = A[1 - e^{-I/I_{\text{sat}}}] + BI \quad (1)$$

The first term of the sum represents the saturating part due to the single exciton component while the second is due to biexciton emission contribution.^{9,28,29} Specifically I_{sat} is the saturation intensity while A and B depend on the intensity of single and biexciton components of the emission.

A typical saturation curve is shown in Figure 2b. To minimize the effect of the blinking on the data analysis, multiple measurements are taken for each experimental intensity in the graph, and only the one with the strongest emission is kept. We observed a median I_{sat} of 0.56 W/ μm^2 . The presence of a biexciton component indicates that the correlation function $g^{(2)}(0)$ will depend on the intensity at which we excite the emitters. We clearly observed this behavior when repeating the measurements for several excitation intensities. To perform reliable measurements on several emitters and to ensure a signal well above the noise level, we performed all the measurements at the saturation intensity.

Effect of the Sample Dilution on the Photobleaching.

Usually perovskite nanocrystals at room temperature suffer from fast photobleaching when they are exposed to light. Several studies presented lead halide perovskites NCs emission instability.³⁰ Often, the monitoring of the spectral stability over few minutes is used to evaluate the photostability,^{9,29} showing that the CEW shifts to more than 10 nm after few tens of seconds for perovskite nanocrystals directly deposited on a glass-plate. Even when they are encapsulated with polystyrene,

the longest measurement time recorded in the literature by Rainò et al.⁹ was around 100 s. The observed spectral drift has been attributed to the degradation of their sizes. In our case, the study of the photostability of the sample synthesized with the procedure described above shows a significant improvement of the photostability of the emitters under illumination. We prepared six samples using different dilutions starting from the most concentrated solution (typical molar concentration is about 1 μM), up to a dilution of 1:50, using toluene as solvent. We started by analyzing the highest concentration sample. With this concentration value, we are still able to individually address each emitter and to collect its luminescence for more than 1 h with strongly reduced bleaching effects. This can be clearly seen in Figure 4a: an emitter is excited for 2 h while its

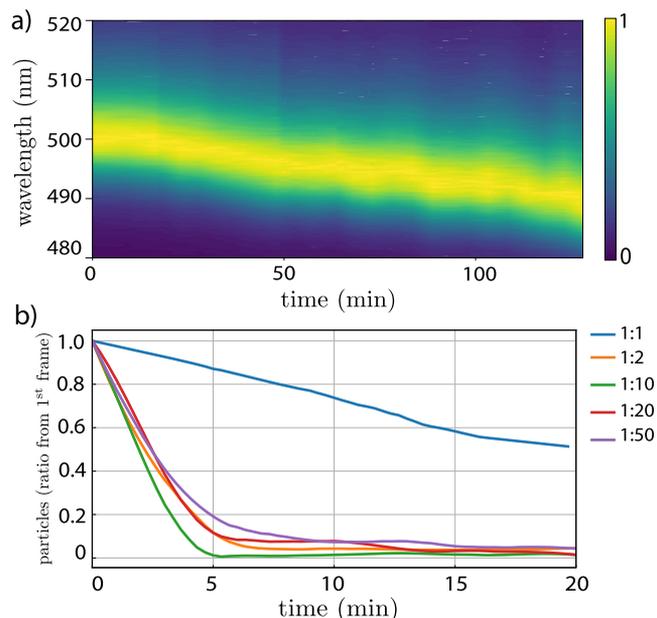


Figure 4. Robustness of perovskites NCs. (a) Evolution vs time of the emission wavelength of a single perovskite NC excited at the saturation intensity. Each spectrum is normalized. (b) Percentage of NCs still emitting after a certain time under strong illumination as a function of the dilution of the original solution: 1:1, 1:2, 1:10, 1:20, 1:50 where 1:X means x times dilution.

emission spectrum is collected every 5 min. The figure reports the evolution of the normalized emission spectra versus time. A blue-shift is observed, as already reported by Rainò et al.,⁹ but on a much longer time scale: indeed our emitters exhibit a blue-shift of less than 10 nm after 2 h, showing a remarkable stability, 2 orders of magnitude better than previously observed.^{5,9} To the best of our knowledge, it is one of the most robust samples reported in the literature.

This robustness is strongly related to the concentration of the emitters in the solution, and drops fast when we dilute the sample. Our final objective is to couple a single emitter to a tapered optical nanofiber to develop an integrated single photon source. For this, we need to use a strongly diluted sample. It is then crucial to investigate the behavior of the emitters as a function of the concentration.

The results of a systematic study of the effect of the dilution are reported in Figure 4b where each sample is strongly illuminated with light from the LED lamp while a video of the sample emission is recorded. Analyzing each frame of the video

(we take a frame each 20 s) we can estimate the number of NCs that are still emitting from the first frame. The important information conveyed by Figure 4b is contained in the initial slope of the curve, which gives information on the characteristic lifetime of an emitter under illumination. We can clearly see that only the concentrated sample (1:1 solution) has a high photostability with more than half of the emitters still working after 20 min, while for the other samples (dilutions from 1:2 to 1:50) almost all the emitters have bleached after 5 min. We observed an increasing number of aggregates for increasing dilutions, indicating that the particles are not colloidally stable anymore. Although our analysis filters out most of the clusters, the applied postselection cannot fully eliminate the smallest aggregates. The residual presence of these aggregates, which last for longer time, results in an overestimation of the number of particles still active after a given time and explains the fact that the 1:50 solution appears better than the 1:2 one.

We attribute this effect to the dynamic bonding of the ligand to the perovskite nanocrystal surface.²¹ Under dilute conditions, the free ligands can hardly find the surface of another nanocrystal. This displaces the equilibrium between bound and unbound ligands toward the latter. As a result, the dilution process leads to poorly passivated nanocrystals which can easily bleach.

Blinking Characterization. The emitted intensity of a NC is not constant in time but tends to fluctuate: this phenomenon is known as blinking. This behavior has been reported for several kinds of quantum emitters such as single molecules,^{31,32} Si nanocrystals,³³ and CdSe/CdS colloidal quantum dots.^{34–39}

Usually the fluorescence blinking is attributed to the trapping of charge carriers. In particular, it has been shown that we can distinguish two types of blinking: Type A, in which the core is left effectively charged and the low fluorescence state is caused by the recombination due to the Auger effect, and Type B, in which the trapped charge can recombine nonradiatively with its opposite charge and the blinking is due to fluctuations in the trapping rate.

A well-established method to experimentally distinguish between Type A and Type B blinking is to study the spontaneous emission lifetime dependence on the emission intensity: in Type A blinking the lifetime is expected to depend on the emission intensity⁴⁰ while in Type B the lifetime should not depend on the intensity.^{40,41}

When the typical blinking time is too short with respect to the chosen binning of the time-trace curve, it becomes impossible to completely distinguish between gray and bright states: in this case it is more appropriate to describe the behavior of the emitter in terms of flickering.⁴⁰

Our emitters, like the vast majority of perovskite NCs reported in the literature,^{42,43} show a clear flickering behavior in their emission time trace. A zoom of a typical blinking trace (i.e., the emitted intensity as a function of time) of one emitter is shown in the upper box of Figure 5a. The signal is binned with a binning time of 50 ms. The complete blinking trace is reported in the Supporting Information. Interestingly, the inspection of this trace indicates a reduced blinking with respect to the typical behavior of this kind of perovskites reported in literature.⁵ In the lower box of Figure 5a, the mean lifetime versus time is reported: a clear correlation between the two curves is observable which indicates the presence of a type A blinking for these emitters.

In order to perform a more quantitative analysis, we fit the lifetime histogram with a triple-exponential decay model:

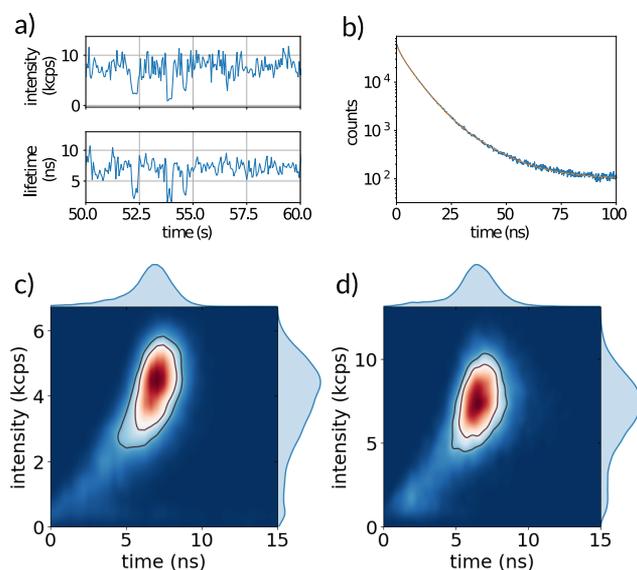


Figure 5. (a) Blinking trace (upper graph) and lifetime (lower graph) of a single NC. (b) Lifetime of a single NC, fitted with a triple-exponential decay model. We obtain 1.4, 6.1, and 14.7 ns, corresponding respectively to the lifetimes of the biexciton, gray and neutral emission states. (c,d) Fluorescence lifetime-intensity distribution (FLID) images of a single emitter, excited at half (c) and twice (d) the saturation intensity. The closed curves contain 50% (inner one) and 68% (outer one) of the occurrences.

$$A_1 e^{-(t-t_0)/\tau_1} + A_2 e^{-(t-t_0)/\tau_2} + A_3 e^{-(t-t_0)/\tau_3} + B \quad (2)$$

with three different lifetimes, τ_1 , τ_2 , and τ_3 , corresponding to neutral, charged, and biexciton state emission, respectively.⁴⁰ In eq 2 t_0 represents the pulse arrival time, while A_i are the amplitudes of each decay component; B is an offset added to take into account the dark counts. A typical result of the fitting procedure is shown in Figure 5b, showing a good agreement with the experimental results. The dependence of the lifetime on the emitted intensity is commonly studied^{5,40,44} using the fluorescence lifetime-intensity distribution (FLID). In Figure 5c and 5d, two FLID images corresponding to the same NC are shown, corresponding to an excitation intensity of $0.5I_{\text{sat}}$ and $2I_{\text{sat}}$ respectively, obtained with a bin size of 50 ms. The dark curves delimit an area corresponding respectively to 50% (inner one) and 68% (outer one) probability of emission. As opposed to previous reports,⁵ showing a predominance of the gray state emission for high excitation powers, with a significant decrease of the emission intensity, remarkably, our perovskite nanocubes remain in the same proportion of bright and gray state while excited up to $2I_{\text{sat}}$ and more, without any significant decrease of the emission intensity.

Quantum Properties. To characterize the quantum emission and verify if our NCs can be used as single photon emitters, we measured the autocorrelation function $g^{(2)}$ using a Hanbury Brown and Twiss (HBT) setup (see Figure 2). The autocorrelation function is defined as follows:

$$g^{(2)}(\tau) = \frac{\langle I(t)I(t+\tau) \rangle}{\langle I(t) \rangle^2}$$

where I is the intensity of the emission, t is the time, and τ is the delay between two different photon arrivals.

Experimentally the beam is divided in two parts by a 50/50 beamsplitter (see Figure 2a) and sent to two avalanche

photodiode single photon detectors (APDs). We record the arrival times with the time tagged time-resolved⁴⁵ method. We then create a histogram of the relative arrival times of the photons with a time bin of 18 ns, shown in Figure 6a. Due to

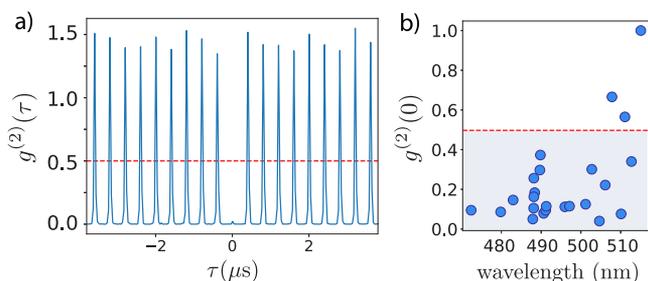


Figure 6. (a) Example of $g^{(2)}$ function of a single NC emitting high quality single photons, measured with a repetition rate of 2.5 MHz. (b) Measured $g^{(2)}(0)$ values as a function of the central wavelength of the emission of the NC. All the emitters were excited at their saturation intensity.

the blinking effect, the peaks close to the 0 delay peak are higher than 1;⁴⁶ on the other hand the peaks of the autocorrelation function tend to 1 for large delays compared to the characteristics blinking time. The histogram is thus normalized by setting the mean height of the peaks with $\tau \approx 10 \mu\text{s}$ to be 1. This procedure is well-established and documented in the literature.^{47,48} The background, mainly given by the dark counts of the APDs, is subtracted following the procedure described in the Supporting Information.

The $g^{(2)}$ at $\tau = 0$ delay measured at saturation is 0.02 according to Figure 6a, well below 0.5 showing the clear signature of a single photon emission. We found that 50% of our emitters show a very good single photon emission, with $g^{(2)}(\tau) < 0.2$. This result shows a single photon emission quality comparable with other room temperature single photons emitters such as CdS/CdSe dot in rod,⁴⁹ InGaN quantum dots,⁵⁰ and NV defects in nanodiamonds.⁵¹

We performed a statistical analysis of the $g^{(2)}(0)$ as a function of the central emission wavelength, shown in Figure 6b. We observe a clear degradation of the single photon emission for longer wavelengths. In particular, for CEWs above 505 nm, the autocorrelation function value is always higher than 0.5, indicating that the single photon emission is lost. The reason for that is that for the biggest NCs the quantum confinement is no longer effective and this has the effect to increase the emission wavelength and to reduce the quality of the single photon emission.

NANOFIBER INTEGRATION

In this section, we describe the first realization of the coupling of a single perovskite NC with an optical tapered nanofiber, constituting a prototype of an interesting hybrid nanophotonic device for quantum technologies-oriented applications. Tapered nanofibers are photonic waveguides obtained by stretching a standard optical fiber while heating it, in order to reduce its diameter to some hundreds of nanometers. This results in a strong evanescent field in the vicinity of the fiber,⁵² which enables the coupling of the light emitted by a nanoemitter located nearby directly into the nanofiber thus obtaining a compact and integrated single photon source.⁵³ This approach has been demonstrated with single atoms^{14,15} as well as single colloidal quantum dots^{16–18} or with nano-

diamonds containing single NV⁻ defects.¹⁹ This is of great interest for quantum applications, where nanofiber-based systems are rapidly developing.⁵⁴

In order to couple our NCs to the nanofiber, we place on top of it a 20 μL droplet with a dilution 1:100 of the original solution in toluene using a micropipette. We then use a translation stage to carefully approach and eventually touch the nanofiber with the droplet, while monitoring the movement with a microscope. This is a critical step, as the nanofiber can easily break. When successful, this procedure results in several emitters deposited onto the nanofiber.

The setup used for this study is shown in Figure 7. We first select the emitter sending the laser in the fiber and detect the

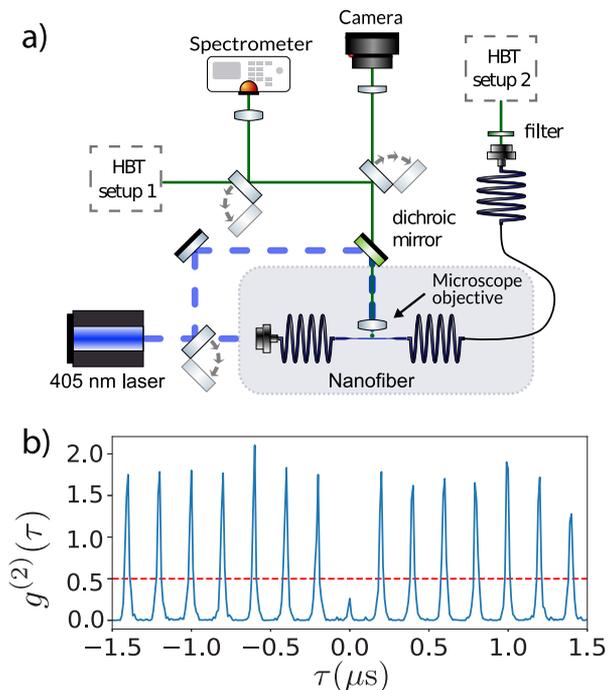


Figure 7. (a) Setup used to measure single photons emission from a NC coupled to the nanofiber. (b) $g^{(2)}$ function of a single NC emitting single photons into the nanofiber (HBT setup 2 of part a), measured with a repetition rate of 5 MHz.

photoluminescence from the microscope objective. Then, we excite a single NC located on top of the nanofiber with a laser sent through that microscope objective. A significant fraction of the emitted light is coupled into the nanofiber and propagates to its end. The end of the fiber is plugged to an HBT setup for a $g^{(2)}$ measurement following the procedure previously described. Optionally, photon antibunching and spectral measurements can be performed also in free-space via the light collected with the microscope objective. The best antibunching measurement performed on the light collected via the nanofiber is shown in Figure 7, corresponding to a $g^{(2)}(0) = 0.24$. While this result is a clear signature of the single photon emission, we did not perform a systematic run of measurements to optimize the value of $g^{(2)}(0)$. For this reason, the observed $g^{(2)}(0)$ is slightly higher than the best values obtained with the coverslip setup. We are thus able to detect single photon emission, and this demonstrates that a single perovskite NC is coupled to a nanofiber and emits single photons directly inside it. With this setup, total collection efficiency up to 30% is expected. These results show that our

NCs are not damaged by the deposition process and validate this approach for this kind of sources, that, due to the dilution, can be used for some minutes once deposited over the nanofiber. Due to the versatility of the perovskite nanocubes and to their emission of single photons with long coherence time,² this technique opens the way to the realization of nanofiber-based, compact, integrated hybrid devices for indistinguishable single photons generation with solid state emitters. In order to achieve this final goal, it is crucial to further improve the stability of the emitters thanks to a deeper understanding of the role that ligands and dilution play on their robustness.

CONCLUSIONS

We investigated the properties of highly stable CsPbBr₃ nanocrystals as single photon emitters. The effect of the dilution on the photostability was investigated, confirming a degradation when increasing the dilution. To further improve the stability, a promising approach, currently under investigation, is based on a better ligand control in order to increase the dilution without breaking the equilibrium of the colloidal solution.

A full characterization of the optical properties of these emitters was performed, highlighting the role of the charge confinement in their antibunching behavior. A deep analysis of the blinking and single photon emission of perovskite nanocubes were presented, showing a strongly reduced blinking and a remarkable stability of the bright state emission as a function of the excitation power. This feature guarantees a very low $g^{(2)}(0)$ also for high excitation power. Moreover, for the first time with such emitters, we have shown the coupling of a single perovskite nanocube with a tapered optical nanofiber. Thanks to the near-field interaction, single photons are emitted in the near field of the nanofiber demonstrating the proof of principle of a compact, integrated single photon source. The coupling to other platforms, such as the ion-integrated waveguides one,⁵⁵ is also envisioned to obtain integrated single photon sources for quantum photonic applications.

EXPERIMENTAL SECTION

Chemicals. PbBr₂ (Alfa Aesar, 98.5%), Cs₂CO₃ (Alfa aesar, 99.99%), oleylamine (OLA, Acros, 80–90%), oleic acid (OA, Sigma-Aldrich), octylamine (oct.Am, Alfa aesar, 99%), octanoic acid (oct.Acid, Acros, 99%), octadecene (ODE, Acros Organics, 90%), toluene (VWR, rechapur).

Cesium Oleate Precursor. We mix in a 50 mL three neck flask, 350 mg of Cs₂CO₃ in 20 mL of ODE and 1.25 mL of OA. The content of the flask is stirred and degassed under vacuum at room temperature for 25 min. The flask is heated at 110 °C for 15 min. The atmosphere is switched to nitrogen and the temperature raised to 150 °C. The reaction is carried on for 15 min. At this stage the salt is fully dissolved. The temperature is cooled down below 100 °C, and the flask is degassed under vacuum. Finally this solution is used as a stock solution.

Nanocrystal Synthesis. In a three neck flask, 174 mg of PbBr₂ is mixed in 10 mL of ODE. The flask is degassed under vacuum at room temperature for 15 min. Then, the temperature is raised to 120 °C. At 105 °C, 0.25 mL of OLA is injected. Once vacuum has recovered, 1.1 mL of OA is injected. After 5 min, 0.75 mL of oct.Am is injected. Once the

vacuum has recovered, 0.75 mL of oct.Acid is injected. The solution is colorless at this stage. The solution is further degassed at 120 °C for 30 min. The atmosphere is switched to nitrogen, and the temperature is raised to 150 °C. Around 0.1 mL of CsOA solution is injected, and the solution turns turbid. The solution is conducted for 35 min and finally quickly cooled down by removing the heating mantle and using a water bath. The solution is transferred to a plastic tube and centrifuged. The supernatant is discarded. The pellet is dispersed in 5 mL of toluene. The solution is centrifuged again at low speed (2000 rpm for 1 min). The pellet is discarded, and the colloiddally stable supernatant is stored.

Material Characterization. UV–visible absorption spectrum are obtained by diluting the NCs in hexane and using a JASCO V730 spectrometer. For transmission electron microscopy, a dilute solution of nanocrystals is drop-casted onto a copper grid coated with a thin amorphous carbon layer. The grid is then degassed under secondary vacuum overnight. Imaging was conducted with a JEOL 2010 microscope operated at 200 kV.

ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsp Photonics.0c00820>.

Sample characterization, integral fluorescence trace, procedure for FLID generation, noise cleaning procedure for $g^{(2)}(\tau)$ measurements, and procedure for NCs deposition on nanofibers (PDF)

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Notes

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